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SOME PROPERTIES OF THE DISSIPATIVE MODEL OF STRAIN-GRADIENT PLASTICITY

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Abstract

A theoretical and computational investigation is carried out of a dissipative model of rate-independent strain-gradient plasticity and its regularization. It is shown that the flow relation, when expressed in terms of the Cauchy stress, is necessarily global. The most convenient approach to formulating the flow relation is through the use of a dissipation function. It is shown, however, that the task of obtaining the dual version, in the form of a normality relation, is a complex one. A numerical investigation of problems in two space dimensions casts further light on the response using the dissipative theory in situations of non-proportional loading. The elastic gap, a feature reported in recent investigations, is observed in situations in which passivation has been imposed. The computational study indicates that the gap may be regarded as an efficient path between a load-deformation response corresponding to micro-free boundary conditions, and that corresponding to micro-hard boundary conditions, in which plastic strains are set equal to zero on all or part of the boundary.

1 Introduction

There has been steady progress in the development of strain-gradient theories of plasticity for over three decades, since the early contribution by Aifantis [1]. The motivation for such theories lies in their ability to capture length-scale dependent effects, which conventional theories are unable to do. Some key works include those by Gao, Huang, Nix and Hutchinson [13, 14], who argue for the inclusion of gradients of plastic strain as a way of accounting for geometrically necessary dislocations, and Fleck and Hutchinson, Gudmundson, and Gurtin and Anand [7, 15, 16], who develop such theoretical models.

This work concerns the small-strain, rate-independent theory of strain-gradient plasticity. The model is based on that proposed for rate-dependent materials by Gurtin and Anand [16] and

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26 **Gudmundson [15]**, and subsequently developed for the rate-independent case by Reddy and
27 coauthors [20, 21], who also carried out an analysis of well-posedness of the problem. The works
28 by Fleck and Willis [10, 11] present and analyze closely related rate-independent and -dependent
29 theories.

30 **In the models referred to above, gradient effects are accounted for either through their inclu-**
31 **sion in the free energy, or in an extension of the flow law, or both.** These first two cases
32 are referred to respectively as energetic and dissipative models, and are both present in many
33 treatments of gradient plasticity. They differ substantially though in their implications for the
34 theory. Fleck, Hutchinson and Willis [8], for example, point out that it is particularly in cases
35 of non-proportional loading that the energetic and dissipative models lead to quite distinct be-
36 haviour. These authors refer to these respectively as incremental and non-incremental theories:
37 their nomenclature stems from the observation that, for energetic (or incremental) theories it is
38 possible to express increments in the microscopic stresses that form part of the description of
39 the model in terms of increments in plastic strain and strain gradients. On the other hand, at
40 least when expressed in local form in a manner that mimics the classical associative flow law,
41 the dissipative model leads to the expression of microscopic stresses – not their increments – in
42 terms of plastic strain and strain gradient increments. These differences in the models are ex-
43 plored and highlighted in [8] in analyses of two problems that involve non-proportional loading.
44 The main distinguishing feature in the two examples is, in the case of the dissipative theory, an
45 elastic gap: that is, elastic behaviour associated with non-proportional loading following loading
46 into the plastic range. This phenomenon has been further investigated by Fleck, Hutchinson
47 and Willis in [9]. **Size effects within a one-dimensional dissipative theory of gradient plasticity**
48 **due to Anand et al. [2] have been analyzed recently by Chiricotto et al. [4].**

49 Fleck and Willis [12] have carried out an analysis for one of the examples studied in detail in
50 [8, 9], viz. the plane-strain tension problem of a strip subjected to passivation of the lateral
51 boundaries at some stage during the loading process. Their analysis shows the existence of the
52 elastic gap for the dissipative problem, and determines theoretically the manner in which plastic
53 flow resumes. A further important outcome of their analysis is the demonstration that the
54 presence of energetic terms has no effect on the size of the elastic gap. **Further recent numerical**
55 **investigations into the appearance of an elastic gap during non-proportional loading include the**
56 **works by Bardella and Panteghini [3, 19].**

57 The yield criterion and associative flow law for the strain-gradient problem gives the plastic
58 strain-rate (or increment) and its gradient in terms of a normality condition that involves the
59 yield function as a function of the microscopic stresses. Unlike the Cauchy stress these are not
60 known a priori in terms of current displacement and plastic strain and therefore cannot be used
61 to determine whether yield has occurred locally, as has been discussed in [10, 11]. It has been
62 shown in [20] however that the microstresses can be eliminated in favour of the Cauchy stress in
63 the flow relation by resorting to a weak or global form of the flow law. This global form is most

64 directly derived and formulated in terms of the dissipation function. The corresponding elastic
65 region, yield function and normality relation can then be derived in principle from a dualization
66 procedure. Knowledge of the current yield surface would in turn provide information relevant
67 to the existence of an elastic gap, whose occurrence must correspond to the stress state lying
68 inside the current elastic region rather than on the yield surface.

69 The objective of this work is to explore various aspects of the dissipative strain-gradient theory,
70 with a view to shedding further light on features of the theory that include those explored in
71 [8, 9, 12]. The global nature of the flow relation, at least when formulated in terms of the
72 Cauchy stress, is shown. The task of determining the corresponding global yield function and
73 normality relation is found not to be readily derivable in closed form. Further insight into
74 the global relation is obtained by analyzing its discrete form, obtained for example from time-
75 discretization combined with finite element approximation in space. Remarkably, even for the
76 discrete problem it is not possible to find in closed form the yield function corresponding to
77 the discrete dissipation function. Rather, an upper bound for the yield function is obtained.
78 These and related issues are examined in further detail in a numerical study using two examples:
79 biaxial deformation of a thin plate, and extension of a circular cylindrical rod. Both are two-
80 dimensional in nature: this allows for an investigation of some features that are not present in the
81 one-dimensional problems in earlier studies. For example, non-proportional loading is effected
82 through the application of passivation, that is, imposition of zero plastic strain increment on
83 part of the boundary, and the resulting yield surface in the multidimensional stress space is
84 explored.

85 An interpretation, from a mathematical perspective, of the elastic gap is given by appealing to
86 the expression for the yield function as a maximum, taken over all admissible plastic strain incre-
87 ments, of a function involving the dissipation. Numerically, the elastic gap appears to constitute
88 an efficient transition from stress-strain behaviour corresponding to a micro-free or Neumann
89 boundary condition, to that which is obtained assuming micro-hard or Dirichlet boundary con-
90 ditions.

91 The plan of the rest of this work is as follows. We summarize the relevant governing relations in
92 Section 2 and derive the flow law in global form, in terms of the dissipation function and involving
93 the Cauchy stress. A mixed formulation, obtained by introducing an auxiliary variable for the
94 plastic strain gradient, is presented in Section 2.2. Section 3 explores the implications of a
95 regularized theory. The dissipation function is not smooth at the origin, and is approximated
96 in Section 3 by one that is smooth. One consequence is that inequalities corresponding to the
97 flow relations are replaced by local or global equations. In Section 4 time-discretization allows
98 the global flow relation to be formulated as one involving plastic strain increments, and for the
99 problem to be formulated as a minimization problem. Such a formulation is not possible for the
100 original problem. In Section 5 we approach the issue of finding the yield function by replacing
101 the original global problem with its fully discrete approximation, and derive an upper bound for

102 the corresponding yield function. Section 6 is devoted to the numerical investigation. Finally,
 103 Section 7 discusses the key results and observations and their implications.

104 2 Governing equations and inequalities

105 The model of strain-gradient plasticity that forms the basis of this study is that proposed by
 106 Gurtin and Anand [16], with the specialization to rate-independent plasticity by Reddy [20].
 107 Small strains are assumed. The displacement is denoted by \mathbf{u} , the total strain by $\boldsymbol{\varepsilon}$, and the
 108 stress by $\boldsymbol{\sigma}$. The strain is decomposed into elastic and plastic components $\boldsymbol{\varepsilon}^e$ and $\boldsymbol{\varepsilon}^p$ according
 109 to

$$\boldsymbol{\varepsilon} = \boldsymbol{\varepsilon}^e + \boldsymbol{\varepsilon}^p. \quad (2.1)$$

110 The strain-gradient theory makes provision for a 2nd-order microscopic stress tensor $\boldsymbol{\pi}$ and
 111 a 3rd-order microscopic stress $\boldsymbol{\Pi}$. The quantity $\boldsymbol{\pi}$ is symmetric and deviatoric, while $\boldsymbol{\Pi}$ is
 112 symmetric and deviatoric in its first two indices, in the sense that $\Pi_{ijk} = \Pi_{jik}$, $\Pi_{ppk} = 0$. Here
 113 and elsewhere the summation convention on repeated indices is invoked, with partial derivatives
 114 denoted by a subscript following a comma.

115 We define the generalized stress \mathbf{S} and plastic strain Γ to be the ordered pairs

$$\mathbf{S} = (\boldsymbol{\pi}, \ell^{-1}\boldsymbol{\Pi}), \quad \Gamma = (\boldsymbol{\varepsilon}^p, \ell\nabla\boldsymbol{\varepsilon}^p). \quad (2.2)$$

116 Here ℓ is a length parameter, and the inner product of the two generalized quantities is denoted
 117 by

$$\mathbf{S} \diamond \Gamma := \boldsymbol{\pi} : \boldsymbol{\varepsilon}^p + \boldsymbol{\Pi} \circ \nabla\boldsymbol{\varepsilon}^p = \pi_{ij}\varepsilon_{ij}^p + \Pi_{ijk}\varepsilon_{ij,k}^p.$$

118 Assuming quasistatic behaviour, the equation of macroscopic equilibrium is given by

$$-\operatorname{div} \boldsymbol{\sigma} = \mathbf{b}, \quad (2.3)$$

119 where \mathbf{b} is the body force. In addition, the stress and microscopic stresses are related to each
 120 other through the microforce balance equation

$$\operatorname{dev} \boldsymbol{\sigma} = \boldsymbol{\pi} - \operatorname{div} \boldsymbol{\Pi} \quad \text{or, in index form,} \quad (\operatorname{dev} \boldsymbol{\sigma})_{ij} = \pi_{ij} - \Pi_{ijk,k}. \quad (2.4)$$

121 Equations (2.3) and (2.4) are required to be satisfied on the domain Ω . The macroscopic
 122 boundary conditions on the problem are

$$\mathbf{u} = \bar{\mathbf{u}} \text{ on } \partial\Omega_u, \quad \boldsymbol{\sigma}\mathbf{n} = \bar{\mathbf{t}} \text{ on } \partial\Omega_t, \quad (2.5)$$

123 where $\partial\Omega_u$ and $\partial\Omega_t$ are complementary parts of the boundary $\partial\Omega$ with unit outward normal
 124 \mathbf{n} , and $\bar{\mathbf{u}}$ and $\bar{\mathbf{t}}$ are respectively a prescribed displacement and surface traction. In addition we

125 assume homogeneous micro-hard and micro-free boundary conditions on complementary parts
 126 $\partial\Omega_H$ and $\partial\Omega_F$ of the boundary; that is,

$$\boldsymbol{\varepsilon}^p = \mathbf{0} \text{ on } \partial\Omega_H, \quad \mathbf{\Pi n} = \mathbf{0} \text{ on } \partial\Omega_F. \quad (2.6)$$

127 Of particular interest is the weak form of the microforce balance equation (2.4). For this purpose
 128 we introduce the space W of plastic strains, defined by

$$W = \{\mathbf{q} \mid q_{ij} = q_{ji}, q_{ii} = 0, q_{ij} \in L^2(\Omega), q_{ij,k} \in L^2(\Omega), q_{ij} = 0 \text{ on } \partial\Omega_H\}.$$

Taking the inner product of (2.4) with arbitrary $\mathbf{q} \in W$, integrating by parts, and imposing the microscopic boundary conditions (2.6), we obtain the weak formulation

$$\begin{aligned} \int_{\Omega} \operatorname{dev} \boldsymbol{\sigma} : \mathbf{q} \, dx &= \int_{\Omega} [\boldsymbol{\pi} : \mathbf{q} + \mathbf{\Pi} \circ \nabla \mathbf{q}] \, dx \\ &= \int_{\Omega} \mathbf{S} \diamond \mathbf{Q} \, dx, \end{aligned} \quad (2.7)$$

129 where $\mathbf{Q} = (\mathbf{q}, \ell \nabla \mathbf{q})$.

130 Given the free energy ψ , the energy imbalance takes the form

$$\dot{\psi} - \boldsymbol{\sigma} : \dot{\boldsymbol{\varepsilon}}^e - \boldsymbol{\pi} : \dot{\boldsymbol{\varepsilon}}^p - \mathbf{\Pi} \circ \nabla \dot{\boldsymbol{\varepsilon}}^p \leq 0. \quad (2.8)$$

131 Since we are concerned in this work with the consequences of a dissipative gradient plasticity
 132 formulation we restrict attention to free energy functions of the form⁶

$$\psi = \psi^e(\boldsymbol{\varepsilon}^e) = \frac{1}{2} \boldsymbol{\varepsilon}^e : \mathbb{C} \boldsymbol{\varepsilon}^e, \quad (2.9)$$

133 in which the elasticity tensor \mathbb{C} is given, for isotropic materials, by

$$\mathbb{C} \boldsymbol{\varepsilon} = \lambda (\operatorname{tr} \boldsymbol{\varepsilon}) \mathbf{I} + 2\mu \boldsymbol{\varepsilon}. \quad (2.10)$$

134 Here λ and μ are the Lamé parameters, and \mathbf{I} is the second-order identity tensor. We note also
 135 for future reference that the deviatoric part of this relation is given by

$$\operatorname{dev} \mathbb{C} \boldsymbol{\varepsilon} = 2\mu \operatorname{dev} \boldsymbol{\varepsilon}. \quad (2.11)$$

136 Substitution of (2.9) in (2.8) and the usual Coleman-Noll procedure lead to the elastic relation

$$\boldsymbol{\sigma} = \frac{\partial \psi^e}{\partial \boldsymbol{\varepsilon}^e} = \mathbb{C} \boldsymbol{\varepsilon}^e \quad (2.12)$$

137 and the reduced dissipation inequality

$$\boldsymbol{\pi} : \dot{\boldsymbol{\varepsilon}}^p + \mathbf{\Pi} \circ \nabla \dot{\boldsymbol{\varepsilon}}^p \geq 0 \quad \text{or} \quad \mathbf{S} \diamond \dot{\boldsymbol{\Gamma}} \geq 0. \quad (2.13)$$

⁶More generally, one considers a free energy that depends in addition on the plastic strain, the plastic strain gradient and, possibly, hardening internal variables. Details may be found, for example, in [20].

138 **2.1 Flow relation**

Based on the reduced dissipation inequality (2.13) we postulate the existence of a yield function f , which is a function of the generalized stress \mathbf{S} , and a flow relation that takes the form of a normality law: that is,

$$f(\mathbf{S}) \leq 0, \tag{2.14a}$$

$$\dot{\Gamma} = \lambda \frac{\partial f}{\partial \mathbf{S}}, \tag{2.14b}$$

$$\lambda \geq 0, \quad f \leq 0, \quad \lambda f = 0. \tag{2.14c}$$

139

Equivalently, as shown schematically in Figure 1,

$$\dot{\Gamma} \diamond (\mathbf{T} - \mathbf{S}) \leq 0 \quad \text{for all } \mathbf{T} \in \mathcal{E} := \{\mathbf{T} \mid f(\mathbf{T}) \leq 0\}, \tag{2.15}$$

140 where \mathcal{E} is the convex elastic region.

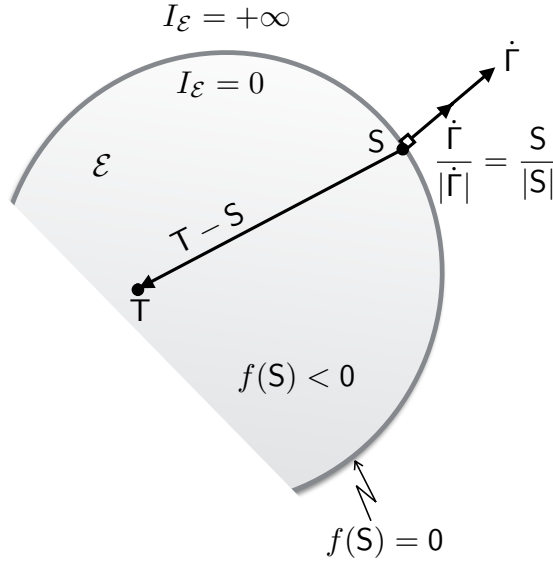


Figure 1: The yield surface and normality relation in generalized stress space

141 The dissipation function D may be defined using a generalization of the postulate of maximum
 142 plastic work in the form

$$D(\dot{\Gamma}) = \sup\{\mathbf{S} \diamond \dot{\Gamma} \mid f(\mathbf{S}) \leq 0\}. \tag{2.16}$$

143 Note that D is convex and positively homogeneous, the latter being defined as $D(\alpha\dot{\Gamma}) = |\alpha|D(\dot{\Gamma})$
 144 for any real number α .

145 **Example** For the special but important case in which

$$f(\mathbf{S}) = |\mathbf{S}| - Y = \sqrt{|\boldsymbol{\pi}|^2 + \ell^{-2}|\mathbf{\Pi}|^2} - Y \leq 0, \quad (2.17)$$

146 where Y is the yield stress, it follows from (2.14b) that at yield ($f = 0$)

$$\lambda = |\dot{\Gamma}| = \sqrt{|\dot{\boldsymbol{\epsilon}}^p|^2 + \ell^2|\nabla\dot{\boldsymbol{\epsilon}}^p|^2}.$$

147 Furthermore, for non-zero $\dot{\Gamma}$,

$$\frac{\dot{\Gamma}}{|\dot{\Gamma}|} = \frac{\mathbf{S}}{|\mathbf{S}|} \iff \mathbf{S} = Y \frac{\dot{\Gamma}}{|\dot{\Gamma}|}. \quad (2.18)$$

148 From (2.16) it is easily seen that for this example

$$D(\dot{\Gamma}) = Y|\dot{\Gamma}|. \quad (2.19)$$

149 □

150 There is an important duality between the flow relation written in terms of the yield and
 151 dissipation functions. To present this we need the notion of the subdifferential ∂F of a convex
 function F , defined here on a finite-dimensional space X such as \mathbb{R}^d (see Figure 2).⁷ Returning

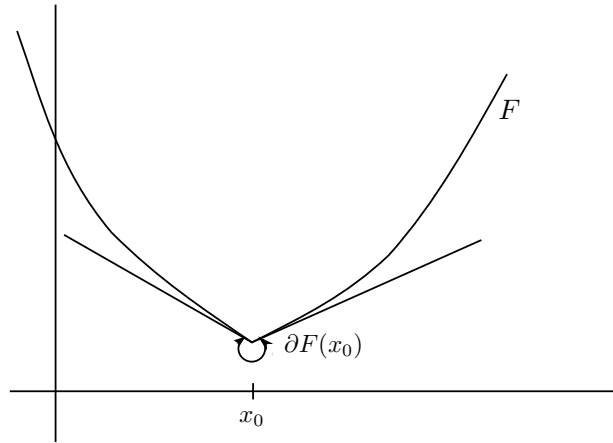


Figure 2: The subdifferential $\partial F(x_0)$ of a convex function F at x_0

152

⁷The subdifferential ∂F of a convex function F is defined by

$$\partial F(\mathbf{x}) = \{\mathbf{p} \mid F(\mathbf{y}) - F(\mathbf{x}) - \mathbf{p} \cdot (\mathbf{y} - \mathbf{x}) \geq 0, \text{ for all } \mathbf{y} \in X\}. \quad (2.20)$$

That is, ∂F is the set of tangents at the point \mathbf{x} . If F is smooth at \mathbf{x} then ∂F comprises a single member, viz. the tangent $\nabla F(\mathbf{x})$ to F at \mathbf{x} , or equivalently the gradient or normal to the level set $F = \text{constant}$. For this and other concepts from convex analysis, see for example [17].

153 to plasticity, we define the indicator function $I_{\mathcal{E}}$ of a set (in this case the elastic region (2.15))
 154 by

$$I_{\mathcal{E}}(\mathbf{S}) = \begin{cases} 0 & \text{if } \mathbf{S} \in \mathcal{E}, \\ +\infty & \text{otherwise.} \end{cases} \quad (2.21)$$

155 This is a convex function. Furthermore, from the definition (2.20) the subdifferential of $I_{\mathcal{E}}$ reads

$$\partial I_{\mathcal{E}}(\mathbf{S}) = \{\dot{\Gamma} \mid \dot{\Gamma} \diamond (\mathbf{T} - \mathbf{S}) \leq 0 \text{ for all } \mathbf{T} \in \mathcal{E}\}. \quad (2.22)$$

156 When compared with (2.15) we see that this is simply the *normality relation*, albeit valid for a
 157 nonsmooth yield function. We use the notation

$$N_{\mathcal{E}}(\mathbf{S}) \quad \text{for} \quad \partial I_{\mathcal{E}}(\mathbf{S}), \quad (2.23)$$

158 given its geometrical interpretation, and refer to $N_{\mathcal{E}}$ as the normal cone to \mathcal{E} at \mathbf{S} . From the
 159 definition $N_{\mathcal{E}} = \{0\}$ if \mathbf{S} lies in the interior (that is, the elastic domain) of \mathcal{E} : as expected, the
 160 generalized plastic strain rate is zero if the generalized stress lies inside the elastic region.

161 From an important result in convex analysis we have the duality relation

$$\dot{\Gamma} \in N_{\mathcal{E}}(\mathbf{S}) \iff \mathbf{S} \in \partial D(\dot{\Gamma}). \quad (2.24)$$

162 The left-hand form of the normality relation has already been established. The equivalence
 163 (2.24) and the definition (2.20) indicate that it may also be written as

$$D(\mathbf{Q}) - D(\dot{\Gamma}) - \mathbf{S} \diamond (\mathbf{Q} - \dot{\Gamma}) \geq 0, \quad (2.25)$$

164 as depicted in Figure 3. If D is differentiable at $\dot{\Gamma}$ then (2.25) reduces to the equation

$$\mathbf{S} = \left. \frac{\partial D}{\partial \mathbf{Q}} \right|_{\mathbf{Q}=\dot{\Gamma}} \quad (2.26)$$

165 (replace \mathbf{Q} by $\dot{\Gamma} \pm \epsilon \mathbf{Q}$ and take the limit $\epsilon \rightarrow 0$).

166 Two important examples of dissipation functions are

$$D_1(\dot{\Gamma}) := Y [|\dot{\epsilon}^p| + \ell |\nabla \dot{\epsilon}^p|] \quad (2.27)$$

167 and

$$D_2(\dot{\Gamma}) := Y |\dot{\Gamma}| = Y \sqrt{|\dot{\epsilon}^p|^2 + \ell^2 |\nabla \dot{\epsilon}^p|^2}. \quad (2.28)$$

168 The function D_2 corresponds to the definition (2.17) of the yield function. For $\dot{\Gamma} \neq 0$, from
 169 (2.26) with $D = D_2$ we recover the relation (2.18)₂. Figure 4 shows the level sets corresponding
 170 to the dissipation functions D_1 and D_2 . It is seen that D_2 is smooth, and so is the corresponding
 171 yield function, while D_1 and its corresponding yield function are piecewise smooth.

172 The dissipation function D_1 is of more than theoretical significance: Evans and Hutchinson
 173 [6] have shown that theories based on such a dissipation give results that correlate well with

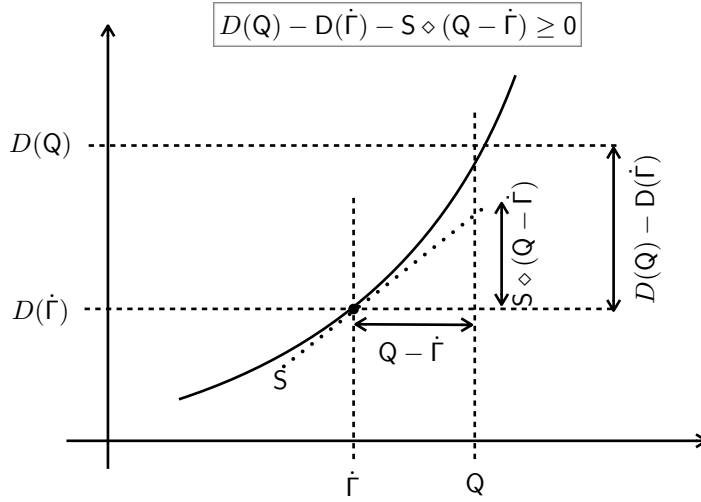


Figure 3: Graphical interpretation of the subdifferential of a convex function D

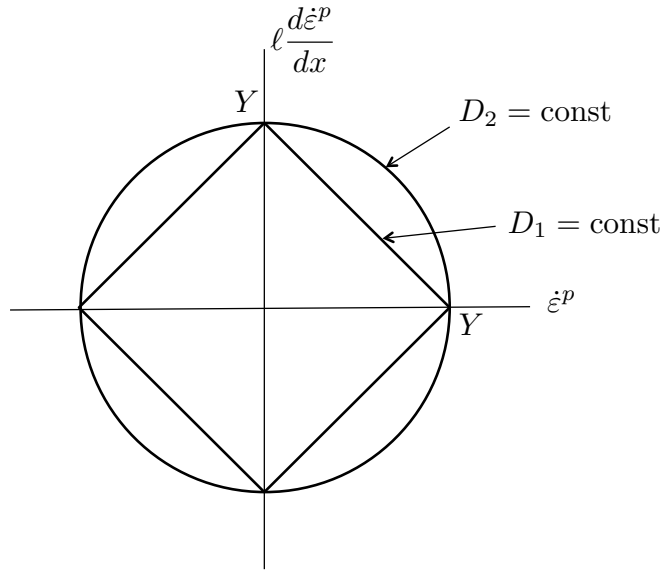


Figure 4: The level sets corresponding to the dissipation functions D_1 and D_2

174 experiments on bending. The yield function corresponding to D_1 is shown in [20] to be piecewise-
 175 smooth or Tresca-like in structure.

176 We now obtain a weak or global form for the flow relation with a view to eliminating S from it.
 177 Integrate (2.25) to obtain

$$\int_{\Omega} [D(Q) - D(\dot{\Gamma}) - S \diamond (Q - \dot{\Gamma})] dx \geq 0 \quad (2.29)$$

178 and add to this the weak form of the microforce balance equation (2.7) to get

$$\int_{\Omega} [D(\mathbf{Q}) - D(\dot{\Gamma}) - \text{dev } \boldsymbol{\sigma} : (\mathbf{q} - \dot{\boldsymbol{\epsilon}}^p)] dx \geq 0 \quad (2.30)$$

179 or

$$\int_{\Omega} [D(\mathbf{Q}) - D(\dot{\Gamma}) - \boldsymbol{\Sigma} \diamond (\mathbf{Q} - \dot{\Gamma})] dx \geq 0, \quad (2.31)$$

180 where

$$\boldsymbol{\Sigma} := (\text{dev } \boldsymbol{\sigma}, \mathbf{0}).$$

181 Concepts such as the subdifferential defined earlier for vectors or tensors at a point (in essence,
182 defined on \mathbb{R}^d) have a broader definition that extends to functionals. Thus, if we define the
183 functional

$$j(\dot{\Gamma}) = \int_{\Omega} D(\dot{\Gamma}) dx, \quad (2.32)$$

184 then the subdifferential of j at $\dot{\Gamma}$ is defined to be the set of functions

$$\partial j(\dot{\Gamma}) = \left\{ \boldsymbol{\Sigma} \mid j(\mathbf{Q}) - j(\dot{\Gamma}) - \int_{\Omega} \boldsymbol{\Sigma} \diamond (\mathbf{Q} - \dot{\Gamma}) dx \geq 0 \right\}. \quad (2.33)$$

185 So we see that (2.31) corresponds to the *global* statement that

$$\boldsymbol{\Sigma} \in \partial j(\dot{\Gamma}). \quad (2.34)$$

186 Furthermore, as in the local case the dual of this relation gives a global normality relation, which
187 we write as

$$\dot{\Gamma} \in \mathcal{N}_{\mathcal{E}_{\text{glob}}}(\boldsymbol{\Sigma}). \quad (2.35)$$

188 The relation (2.35) is equivalent to finding the global form of the normality relation and the
189 corresponding yield function. This is not a trivial task, as we shall see in Section 5.2 where,
190 even for a discrete and therefore finite-dimensional approximation to the problem, at best it is
191 possible to find only an upper bound to the yield function.

192 **Remark** Note that the microstress \mathbf{S} has been eliminated from the global flow relation. This
193 will be important in interpreting the flow relation for the gradient problem, as the local form
194 (2.14) involves \mathbf{S} , which is *indeterminate* in the elastic region.

195 2.2 A mixed formulation for the dissipation function D_1

196 If the dissipation function D were a function of two independent variables, it would be feasible to
197 obtain the corresponding yield function and normality law (2.35) by appealing to standard results
198 from convex analysis. The arguments in D are however the plastic strain and its gradient, and
199 this relationship between the two variables complicates the task of finding the yield condition.
200 With this in mind we explore a mixed approach in which the plastic strain gradient is treated
201 as an independent variable.

202 For convenience we make use of the dissipation function D_1 defined in (2.27), and introduce the
 203 auxiliary variable \mathbf{P} , a third-order tensor defined by

$$\mathbf{P} = \ell \nabla \varepsilon^p. \quad (2.36)$$

The dissipation function is now a function of two independent variables and can be written

$$\begin{aligned} D_1(\varepsilon^p, \mathbf{P}) &:= D_{10}(\varepsilon^p) + D_{01}(\mathbf{P}) \\ &= Y|\varepsilon^p| + Y|\mathbf{P}|. \end{aligned} \quad (2.37)$$

204 The corresponding flow relation reads

$$(\boldsymbol{\pi}, \boldsymbol{\Pi}) \in \partial D_1(\dot{\varepsilon}^p, \dot{\mathbf{P}}). \quad (2.38)$$

Since the two arguments of D_1 are now independent we may use an identity ([5], (Ch. III, eqn (4.17), page 61)) to obtain

$$I_{\mathcal{E}}(\boldsymbol{\pi}, \boldsymbol{\Pi}) = I_{\mathcal{E}_{10}}(\boldsymbol{\pi}) + I_{\mathcal{E}_{01}}(\boldsymbol{\Pi}). \quad (2.39)$$

205 Here $I_{\mathcal{E}}$ is the indicator function for the set \mathcal{E} , $\mathcal{E}_{10} = \{\boldsymbol{\pi} \mid |\boldsymbol{\pi}| \leq Y\}$ and $\mathcal{E}_{01} = \{\boldsymbol{\Pi} \mid |\boldsymbol{\Pi}| \leq Y\}$.
 206 Thus the use of a mixed approach allows the corresponding elastic region to be obtained easily.

207 The flow relation (2.38) is

$$\int_{\Omega} D_1(\mathbf{q}, \mathbf{Q}) dx - \int_{\Omega} D_1(\dot{\varepsilon}^p, \dot{\mathbf{P}}) dx - \int_{\Omega} [\boldsymbol{\pi} : (\mathbf{q} - \dot{\varepsilon}^p) + \boldsymbol{\Pi} \circ (\mathbf{Q} - \dot{\mathbf{P}})] dx \geq 0, \quad (2.40)$$

208 where \mathbf{q} and \mathbf{Q} are respectively an arbitrary plastic strain and auxiliary variable. Set $\mathbf{q} = \mathbf{q} - \dot{\varepsilon}^p$
 209 in (2.7) and add to (2.40) to obtain

$$\int_{\Omega} D_1(\mathbf{q}, \mathbf{Q}) dx - \int_{\Omega} D_1(\dot{\varepsilon}^p, \dot{\mathbf{P}}) dx - \int_{\Omega} [\boldsymbol{\Pi} \circ ((\nabla \mathbf{q} - \mathbf{Q}) - (\nabla \dot{\varepsilon}^p - \dot{\mathbf{P}}))] dx - \int_{\Omega} \text{dev } \boldsymbol{\sigma} : (\mathbf{q} - \dot{\varepsilon}^p) dx \geq 0. \quad (2.41)$$

By setting first \mathbf{Q} , and then \mathbf{q} , equal to zero, we extract the two variational inequalities

$$\int_{\Omega} Y|\mathbf{q}| dx - \int_{\Omega} Y|\dot{\varepsilon}^p| dx - \int_{\Omega} \boldsymbol{\Pi} \circ \nabla(\mathbf{q} - \dot{\varepsilon}^p) dx - \int_{\Omega} \text{dev } \boldsymbol{\sigma} : (\mathbf{q} - \dot{\varepsilon}^p) dx \geq 0, \quad (2.42a)$$

$$\int_{\Omega} Y|\mathbf{Q}| dx - \int_{\Omega} Y|\dot{\mathbf{P}}| dx + \int_{\Omega} \boldsymbol{\Pi} \circ (\mathbf{Q} - \dot{\mathbf{P}}) dx \geq 0. \quad (2.42b)$$

210 To these must be added the weak form of (2.36), that is,

$$\int_{\Omega} \mathbf{P} \circ \mathbf{Q} dx - \int_{\Omega} \ell \nabla \varepsilon^p \circ \mathbf{Q} dx = 0 \quad \text{for all } \mathbf{Q}, \quad (2.43)$$

211 and the weak form of the equilibrium equation (2.3) together with the boundary conditions
 212 (2.5): that is,

$$\int_{\Omega} \boldsymbol{\sigma}(\mathbf{u}, \varepsilon^p) : \boldsymbol{\varepsilon}(\mathbf{v}) dx = \int_{\Omega} \mathbf{b} \cdot \mathbf{v} dx + \int_{\partial \Omega_t} \bar{\mathbf{t}} \cdot \mathbf{v} ds, \quad (2.44)$$

213 in which the test functions \mathbf{v} satisfy the homogeneous boundary condition $\mathbf{v} = \mathbf{0}$ on $\partial \Omega_u$. We
 214 omit details of the (standard) function space setting for the set of weak equations.

215 Equations (2.42), (2.43) and (2.44) constitute a mixed problem for \mathbf{u} , ε^p , \mathbf{P} and $\boldsymbol{\Pi}$. This appears
 216 to be a nonstandard mixed problem.

217 **3 The regularized problem**

218 Later, when developing a computational approach we will focus on the dissipation function D_2 ,
 219 which is an elliptical cone and therefore smooth everywhere except at the origin. It will be
 220 convenient to replace D_2 by a regularized approximation $D_{2\eta}$, defined for $\eta > 0$ by

$$D_{2\eta}(\Gamma) = Y \sqrt{|\boldsymbol{\varepsilon}^p|^2 + \ell^2 |\nabla \boldsymbol{\varepsilon}^p|^2 + \eta^2}. \quad (3.1)$$

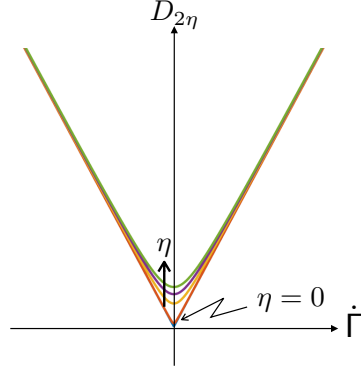


Figure 5: The regularized dissipation function $D_{2\eta}$

221

222 The function is shown in Figure 5. The local flow relation corresponding to $D_{2\eta}$ becomes, with
 223 the use of (2.26),

$$\mathbf{S} = \nabla D_{2\eta}(\dot{\Gamma}) \iff \begin{cases} \boldsymbol{\pi} = \frac{\partial D_{2\eta}}{\partial \mathbf{q}} \Big|_{\mathbf{q}=\dot{\boldsymbol{\varepsilon}}^p} = \frac{Y^2 \dot{\boldsymbol{\varepsilon}}^p}{D_{2\eta}}, \\ \boldsymbol{\Pi} = \frac{\partial D_{2\eta}}{\ell \partial \nabla \mathbf{q}} \Big|_{\nabla \mathbf{q}=\nabla \dot{\boldsymbol{\varepsilon}}^p} = \frac{Y^2 \nabla \dot{\boldsymbol{\varepsilon}}^p}{D_{2\eta}}. \end{cases} \quad (3.2)$$

224 Moreover, the inequality (2.29) becomes the equation

$$\int_{\Omega} [\nabla D_{2\eta}(\dot{\Gamma}) - \boldsymbol{\Sigma}] \diamond \mathbf{Q} \, dx = 0, \quad (3.3)$$

225 OR

$$\int_{\Omega} \left[\left(\frac{Y^2}{D_{2\eta}} \dot{\boldsymbol{\varepsilon}}^p - \operatorname{dev} \boldsymbol{\sigma} \right) : \mathbf{q} + \frac{Y^2}{D_{2\eta}} \ell^2 \nabla \dot{\boldsymbol{\varepsilon}}^p \circ \nabla \mathbf{q} \right] dx = 0. \quad (3.4)$$

226 Assuming sufficient smoothness, integrating by parts and making use of the boundary conditions
 227 (2.6), we obtain the weak equation

$$\int_{\Omega} \left[\left(\frac{Y^2}{D_{2\eta}} \dot{\boldsymbol{\varepsilon}}^p - \operatorname{dev} \boldsymbol{\sigma} \right) - \ell^2 \operatorname{div} \left(\frac{Y^2}{D_{2\eta}} \nabla \dot{\boldsymbol{\varepsilon}}^p \right) \right] : \mathbf{q} \, dx = 0. \quad (3.5)$$

228 This leads to the pointwise relation

$$\operatorname{dev} \boldsymbol{\sigma} = \frac{Y^2}{D_{2\eta}} \dot{\boldsymbol{\varepsilon}}^p - \ell^2 \operatorname{div} \left(\frac{Y^2 \nabla \dot{\boldsymbol{\varepsilon}}^p}{D_{2\eta}} \right). \quad (3.6)$$

229 We note that $(\operatorname{div} \nabla \dot{\boldsymbol{\varepsilon}}^p)_{ij} = \dot{\varepsilon}_{ij,kk}^p$ and that this quantity is deviatoric if $\dot{\boldsymbol{\varepsilon}}^p$ is. Equation (3.6),
 230 which will form the basis of the computational investigation reported in Section 6, could have
 231 been obtained directly by substituting the regularized version of the flow relation (3.2) in the
 232 microforce balance equation (2.4). Indeed, the first and second terms on the right-hand side of
 233 (3.6) correspond respectively to $\boldsymbol{\pi}$ and $-\operatorname{div} \boldsymbol{\Pi}$.

234 4 A time-discrete minimization problem

235 The global problem (2.31) does not have an equivalent minimization problem. However, the
 236 corresponding time-discrete problem may be posed as a minimization problem. We discretize
 237 in time by partitioning the time interval $[0, T]$ as $0 = t_1 < t_2 < \dots < t_n < \dots < t_N = T$, set
 238 $w_n := w(t_n)$ and $\Delta w = w_{n+1} - w_n$ for any function w , and replace the time derivative \dot{w} by its
 239 backward Euler approximation $\Delta w / \Delta t$. Then (2.31) becomes

$$\int_{\Omega} [D(\mathbf{Q}) - D(\Delta \Gamma) - \Sigma_{n+1} \diamond (\mathbf{Q} - \Delta \Gamma)] dx \geq 0. \quad (4.1)$$

Here we have multiplied throughout by Δt , made use of the positive homogeneity of D , and
 replaced the arbitrary $\mathbf{Q} \Delta t$ by \mathbf{Q} . Now from (2.11), and noting that $\Delta \boldsymbol{\varepsilon}^p$ is deviatoric,

$$\begin{aligned} \operatorname{dev} \boldsymbol{\sigma}_{n+1} &= \operatorname{dev} [\mathbb{C}(\boldsymbol{\varepsilon}_{n+1} - \boldsymbol{\varepsilon}_n^p - \Delta \boldsymbol{\varepsilon}^p)] \\ &= \boldsymbol{\sigma}^{\operatorname{tr}} - 2\mu \Delta \boldsymbol{\varepsilon}^p, \end{aligned} \quad (4.2)$$

where

$$\begin{aligned} \boldsymbol{\sigma}^{\operatorname{tr}} &:= \operatorname{dev} [\mathbb{C}(\boldsymbol{\varepsilon}_{n+1} - \boldsymbol{\varepsilon}_n^p)] \\ &= 2\mu (\operatorname{dev} \boldsymbol{\varepsilon}_{n+1} - \boldsymbol{\varepsilon}_n^p) \end{aligned} \quad (4.3)$$

240 is a deviatoric trial elastic stress; that is, the deviatoric stress corresponding to purely elastic
 241 behaviour in the time step $t_n \rightarrow t_{n+1}$. Thus (4.1) becomes

$$\int_{\Omega} [D(\mathbf{Q}) - D(\Delta \Gamma) + 2\mu \Delta \boldsymbol{\varepsilon}^p : (\mathbf{q} - \Delta \boldsymbol{\varepsilon}^p) - \boldsymbol{\sigma}^{\operatorname{tr}} : (\mathbf{q} - \Delta \boldsymbol{\varepsilon}^p)] dx \geq 0. \quad (4.4)$$

242 This is equivalent to the minimization problem

$$\Delta \boldsymbol{\varepsilon}^p = \operatorname{argmin}_{\mathbf{Q}} L(\mathbf{Q}), \quad (4.5)$$

243 where

$$L(\mathbf{Q}) := \int_{\Omega} [D(\mathbf{Q}) + \mu |\mathbf{q}|^2 - \boldsymbol{\sigma}^{\operatorname{tr}} : \mathbf{q}] dx, \quad (4.6)$$

244 for given $\boldsymbol{\sigma}^{\operatorname{tr}}$ where as before $\mathbf{Q} = (\mathbf{q}, \ell \nabla \mathbf{q})$. Note that, unlike the classical case, this is a global
 245 problem which cannot be reduced to a local or pointwise one, given that \mathbf{Q} involves \mathbf{q} and its
 246 gradient.

247 5 The spatially discrete problem

248 5.1 Discrete flow relations

249 In this section we examine features of the spatially discrete problem. We also discretize in time
 250 as in Section 4. The domain, assumed polygonal (in two dimensions) or polyhedral (in three)
 251 for convenience, is covered by a mesh comprising

$$\text{NE elements and NN nodes} \quad (5.1)$$

252 where NN excludes those nodes at which the plastic strain is prescribed. The number of plastic
 253 strain degrees of freedom at each node is, taking into account the symmetry of $\boldsymbol{\varepsilon}^p$ and the plastic
 254 incompressibility condition $\text{tr } \boldsymbol{\varepsilon}^p = 0$,

$$\text{ndofs} = d(d+1)/2 - 1 \quad (5.2)$$

255 for a d -dimensional problem ($d > 1$).

256 Denote the global degrees of freedom of $\boldsymbol{\varepsilon}^p$ by \mathbf{p} and those of the displacement by \mathbf{d} , and assume
 257 conventional conforming approximations. Then

$$\boldsymbol{\varepsilon}^p = \mathbf{N}\mathbf{p}, \quad \nabla \boldsymbol{\varepsilon}^p = \mathbf{B}\mathbf{p}, \quad \mathbf{u} = \bar{\mathbf{N}}\mathbf{d}, \quad \boldsymbol{\varepsilon}(\mathbf{u}) = \bar{\mathbf{B}}\mathbf{d}, \quad (5.3)$$

258 where \mathbf{N} and $\bar{\mathbf{N}}$ are matrices of shape functions and \mathbf{B} and $\bar{\mathbf{B}}$ matrices of shape function deriva-
 259 tives.

260 Here and elsewhere we drop the subscript n that denotes quantities at time t_n .

261 Since

$$|\boldsymbol{\varepsilon}^p| = \sqrt{\mathbf{p}^T \mathbf{N}^T \mathbf{N} \mathbf{p}} \quad \text{and} \quad |\nabla \boldsymbol{\varepsilon}^p| = \sqrt{\mathbf{p}^T \mathbf{B}^T \mathbf{B} \mathbf{p}},$$

262 we have, from (2.27),

$$D_1(\mathbf{p}) = Y \left[\sqrt{\mathbf{p}^T \mathbf{N}^T \mathbf{N} \mathbf{p}} + \ell \sqrt{\mathbf{p}^T \mathbf{B}^T \mathbf{B} \mathbf{p}} \right], \quad (5.4)$$

which is homogeneous of degree 1 in \mathbf{p} . Likewise,

$$\begin{aligned} D_2(\mathbf{p}) &= Y \left[\sqrt{\mathbf{p}^T \mathbf{N}^T \mathbf{N} \mathbf{p} + \ell^2 \mathbf{p}^T \mathbf{B}^T \mathbf{B} \mathbf{p}} \right] \\ &= Y \sqrt{\mathbf{p}^T \mathbf{K} \mathbf{p}}, \end{aligned} \quad (5.5)$$

263 where the pointwise matrix \mathbf{K} is defined by

$$\mathbf{K} = \mathbf{N}^T \mathbf{N} + \ell^2 \mathbf{B}^T \mathbf{B}.$$

264 Next, set

$$\mathcal{J}_i(\mathbf{q}) := \int_{\Omega} D_i(\mathbf{q}) \, dx \quad (i = 1, 2); \quad (5.6)$$

then (2.30) becomes, for the incremental problem,

$$\mathcal{J}_i(\mathbf{q}) - \mathcal{J}_i(\Delta\mathbf{p}) - (\mathbf{q} - \Delta\mathbf{p})^T \mathbf{s} \geq 0 \quad \text{for all } \mathbf{q}, \quad (5.7)$$

265 where the global vector of nodal stresses \mathbf{s} is given by

$$\mathbf{s} := \int_{\Omega} \mathbf{N}^T \text{dev } \boldsymbol{\sigma} \, dx. \quad (5.8)$$

266 Thus we have the discrete inclusion

$$\mathbf{s} \in \partial\mathcal{J}_i(\Delta\mathbf{p}) \quad (5.9)$$

and the dual of this is, from (2.24),

$$\Delta\mathbf{p} \in N_{\mathcal{E}_i}(\mathbf{s}) \quad \text{or} \quad (\Delta\mathbf{p})^T (\mathbf{t} - \mathbf{s}) \leq 0 \quad \text{for all } \mathbf{t} \in \mathcal{E}_i, \quad (5.10)$$

267 in which \mathcal{E}_i is the elastic region, in the space of discrete stresses \mathbf{s} , corresponding to the dissipation
 268 function \mathcal{J}_i . As with the continuous problem the inclusion (5.9) is equivalent to a minimization
 269 problem. First, we have

$$\mathbf{s} = \underbrace{\int_{\Omega} \mathbf{N}^T \boldsymbol{\sigma}^{\text{tr}} \, dx}_{\mathbf{s}^{\text{tr}}} - \underbrace{\int_{\Omega} 2\mu \mathbf{N}^T \mathbf{N} \, dx}_{\mathbf{M}} \Delta\mathbf{p}, \quad (5.11)$$

270 so that the minimization problem is

$$\Delta\mathbf{p} = \operatorname{argmin}_{\mathbf{q}} (\mathcal{J}_i(\mathbf{q}) + \mathbf{q}^T \mathbf{M} \mathbf{q} - \mathbf{q}^T \mathbf{s}^{\text{tr}}). \quad (5.12)$$

271 Thus we have obtained a vehicle to establish the relation between the dissipation function
 272 corresponding to the global dissipation functions \mathcal{J}_i and their corresponding elastic regions \mathcal{E}_i .

273 5.2 Finding the elastic region

We have available the global dissipation functions \mathcal{J}_i and now seek to construct the corresponding elastic regions \mathcal{E}_i and associated yield functions, which would allow the use of the flow law as a normality relation, as in (5.10). Now from a result in convex analysis (see for example [17], page 109), given a dissipation function \mathcal{J} , one may construct a yield function $\phi(\mathbf{s})$ as a function of the global nodal stresses with the properties

$$\phi \text{ is positively homogeneous and convex,} \quad (5.13a)$$

$$\mathcal{E} = \{\mathbf{s} \mid \phi(\mathbf{s}) \leq 1\}, \quad (5.13b)$$

$$\phi(\mathbf{s}) = \sup_{\mathbf{q} \neq 0} \frac{\mathbf{q}^T \mathbf{s}}{\mathcal{J}(\mathbf{q})}. \quad (5.13c)$$

274 It follows that the yield function can be constructed if we are able to evaluate the supremum in
 275 (5.13c).

276 Locally, the relationship (5.13c) is exemplified in the yield and dissipation functions (2.17) and
 277 (2.28). Unfortunately, determining ϕ in (5.13c) corresponding to the *global* functions \mathcal{J}_i is not
 278 a simple task, as will be seen: the best that can be done is to obtain a function that is an upper
 279 bound for ϕ . To see this, we focus on the dissipation function \mathcal{J}_2 and note that this can be
 280 written, for constant yield stress Y , as

$$\mathcal{J}_2(\mathbf{q}) = Y \int_{\Omega} |\mathbf{K}^{1/2} \mathbf{q}| \, dx .$$

Taking $|\Omega| = 1$ for convenience we note that

$$\begin{aligned} \mathbf{s}^T \mathbf{q} &= \int_{\Omega} ([\mathbf{K}(\mathbf{x})]^{-1/2} \mathbf{s})^T ([\mathbf{K}(\mathbf{x})]^{1/2} \mathbf{q}) \, dx \\ &\leq \int_{\Omega} |[\mathbf{K}(\mathbf{x})]^{-1/2} \mathbf{s}| |[\mathbf{K}(\mathbf{x})]^{1/2} \mathbf{q}| \, dx \\ &\leq Y^{-1} (\max_{\mathbf{x} \in \Omega} |[\mathbf{K}(\mathbf{x})]^{-1/2} \mathbf{s}|) \mathcal{J}_2(\mathbf{q}) . \end{aligned} \quad (5.14)$$

Hence we have, from (5.13c) and (5.14),

$$\begin{aligned} \phi(\mathbf{s}) &= \sup_{\mathbf{q} \neq 0} \frac{\mathbf{q}^T \mathbf{s}}{Y \int_{\Omega} |\mathbf{K}^{1/2}(\mathbf{x}) \mathbf{q}| \, dx} \\ &\leq Y^{-1} \max_{\mathbf{x} \in \Omega} |[\mathbf{K}(\mathbf{x})]^{-1/2} \mathbf{s}| . \end{aligned} \quad (5.15)$$

281 We shall now show that the upper bound in (5.15) is not achieved. In order for the expression on
 282 the right-hand side of (5.15) to be equivalent to the yield function ϕ , the supremum in the first
 283 line of (5.15) has to be achieved at this value. That is, assuming the supremum to be achieved
 284 for $\bar{\mathbf{q}} \neq 0$, we must have

$$\frac{\bar{\mathbf{q}}^T \mathbf{s}}{\int_{\Omega} |\mathbf{K}^{1/2}(\mathbf{x}) \bar{\mathbf{q}}| \, dx} = \max_{\mathbf{x} \in \Omega} |[\mathbf{K}(\mathbf{x})]^{-1/2} \mathbf{s}| ,$$

285 OR

$$\frac{\int_{\Omega} ([\mathbf{K}(\mathbf{x})]^{-1/2} \mathbf{s})^T ([\mathbf{K}(\mathbf{x})]^{1/2} \bar{\mathbf{q}}) \, dx}{\int_{\Omega} |\mathbf{K}^{1/2}(\mathbf{x}) \bar{\mathbf{q}}| \, dx} = \max_{\mathbf{x} \in \Omega} |[\mathbf{K}(\mathbf{x})]^{-1/2} \mathbf{s}| .$$

286 Since this equation must hold for any \mathbf{s} , we require that \mathbf{K} be constant, which is a contradiction.

287 **5.3 Illustration of the theory using the problem of a strip in tension**

288 In order to contextualise the theory and to elucidate the numerical results presented in Section
 289 6, we briefly review the study by Fleck, Hutchinson and Willis [8, 9] of a strip in tension that is
 290 subjected to passivation on two lateral surfaces at a certain point in its loading history.

291 The problem is one in plane strain, of a strip $(-\infty \times \infty) \times (-h, h)$ that is subjected to a uniform
 292 applied strain $\varepsilon_{11} \equiv \bar{\varepsilon}$ in the x -direction. The only non-zero plastic strain components are ε_{11}^p
 293 and $\varepsilon_{22}^p = -\varepsilon_{11}^p$, which follows from plastic incompressibility.

294 The surfaces $y = \pm h$ are initially traction-free and micro-traction free. At a certain point in the
 295 loading history beyond that of initial yield these surfaces are passivated, resulting in the plastic
 296 strain rate being zero on the boundaries from this point onwards. The authors in [8] report an
 297 *elastic gap*: that is, purely elastic behaviour following passivation, with plastic flow occurring
 298 after the load has increased somewhat.

299 The strain $\varepsilon_{11} = \bar{\varepsilon}$ is prescribed and increases monotonically. We therefore use the time t as a
 300 parameter.

301 The problem in question is one-dimensional, so for definiteness consider a mesh of uniform 1D
 302 elements with nodes 1 - 5 located respectively at $x = 0$ and $y = h, h/2, 0, -h/2, -h$ (Figure 6).
 303 From symmetry $\mathbf{p}_2 = \mathbf{p}_4$ and $\mathbf{p}_1 = \mathbf{p}_5$.

304 *The uniform phase*

305 For time steps t_1, t_2, \dots, t_n , $\mathbf{p}_1 = \mathbf{p}_5 \neq \mathbf{0}$ and

$$(\mathbf{p}_i)_n \equiv \bar{\mathbf{p}} \quad (i = 1, \dots, 5). \quad (5.16)$$

Assume that the yield stress Y is given. For the case of uniform deformation $\nabla \varepsilon^p = \mathbf{0}$ and the

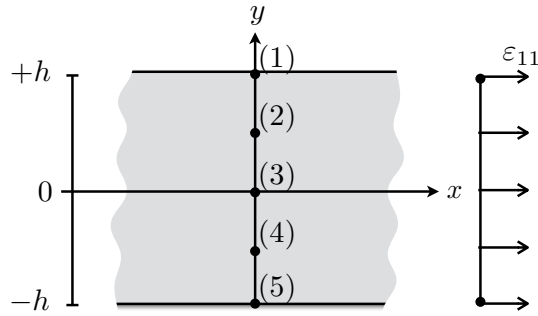


Figure 6: Finite element mesh for the problem of a strip in uniform tension

306
 307 dissipation and yield functions are the conventional ones.

308 *The passivation phase*

309 Consider the first time step following passivation: we now have $\Delta \mathbf{p}_1 = \Delta \mathbf{p}_5 = \mathbf{0}$ and there are
 310 only three free degrees of freedom, corresponding to nodes 2, 3, 4. Each $\Delta \mathbf{p}_i$ has one independent
 311 component since there is no shear, and from plastic incompressibility $\varepsilon_{22}^p = -\varepsilon_{11}^p$.

Some insight into the elastic gap may be gained by making use of the definition (5.13c) of the
 canonical yield function ϕ . Denote by N_{unif} the number of nodal degrees of freedom of plastic

strain: this will be equal to the total number of nodes, since the boundary condition is micro-traction free. Likewise, denote by N_{pass} the number of degrees of freedom in the passivation phase. We have $N_{\text{pass}} < N_{\text{unif}}$ as the plastic strain increment is prescribed to be zero on the boundary nodes. Assuming for convenience that the difference between the vectors \mathbf{s} of nodal stresses just before and after initiation of passivation is negligible, it follows from (5.13c) that

$$\begin{aligned}\phi_{\text{pass}}(\mathbf{s}) &= \sup \frac{\mathbf{s}^T \mathbf{q}_{\text{pass}}}{\mathcal{J}(\mathbf{q}_{\text{pass}})} \\ &\leq \sup \frac{\mathbf{s}^T \mathbf{q}_{\text{unif}}}{\mathcal{J}(\mathbf{q}_{\text{unif}})} \\ &= \phi(\mathbf{s}_{\text{unif}}) = 1.\end{aligned}\tag{5.17}$$

312 Here we have denoted by \mathbf{q}_{unif} and \mathbf{q}_{pass} arbitrary vectors in the uniform and passivated phases,
313 respectively. Also, in the last line we use the assumption that the material is in the plastic range
314 in the uniform phase just before passivation. The inequality in the second line follows from the
315 fact that $\dim \mathbf{q}_{\text{pass}} = N_{\text{pass}} < N_{\text{unif}} = \dim \mathbf{q}_{\text{unif}}$, so that the supremum is being taken over a
316 larger set. From this bound it is clearly possible that $\phi_{\text{pass}} < 1$, so that the response *could be*
317 *elastic* in the initial passivation phase. That this is the case for this particular problem has been
318 shown theoretically in [12].

When plastic flow does eventually take place, the inclusion (5.9) gives an explicit expression for the stress, viz.

$$\begin{aligned}\mathbf{s}_{n+1} &= \left. \frac{\partial \mathcal{D}}{\partial \mathbf{p}} \right|_{\Delta \mathbf{p}} \\ &= \int_{\Omega} \frac{\mathbf{K} \Delta \mathbf{p}}{\sqrt{(\Delta \mathbf{p})^T \mathbf{K} \Delta \mathbf{p}}} dx.\end{aligned}\tag{5.18}$$

319 Unlike the classical case, it appears not to be possible to obtain a closed-form expression for the
320 inverted form of this expression.

321 6 Numerical investigation

We consider fully discrete approximations of the problem, based on weak forms of the equilibrium, microforce balance, and flow relations, with time-discretization as set out in Section 4, and making use of the regularized form (3.1) of the dissipation function. Assume that the state of the system is known at time t_n and that a backward Euler time-integration scheme is employed. The weak form of the equilibrium and microforce balances (2.3)–(2.4) for the system at $t_{n+1} = t_n + \Delta t$ (the system of residual equations) are given by

$$R_d := \int_{\Omega} \boldsymbol{\varepsilon}(\mathbf{v}) : \boldsymbol{\sigma}_{n+1} dx - \int_{\partial \Omega_t} \mathbf{v} \cdot \bar{\mathbf{t}}_{n+1} ds,\tag{6.1}$$

$$R_p := \int_{\Omega} \mathbf{q} : \text{dev } \boldsymbol{\sigma}_{n+1} dx - \int_{\Omega} \mathbf{q} : \boldsymbol{\pi}_{n+1} dx - \int_{\Omega} \nabla \mathbf{q} : \mathbf{\Pi}_{n+1} dx,\tag{6.2}$$

where as before \mathbf{v} and \mathbf{q} are respectively displacement and plastic strain test functions, $\boldsymbol{\pi}$ and $\mathbf{\Pi}$ are given by (3.2), and

$$\boldsymbol{\sigma}_{n+1} = \mathbb{C}(\boldsymbol{\varepsilon}(\mathbf{u}_{n+1}) - \boldsymbol{\varepsilon}_{n+1}^p).$$

322 For convenience we make use of the regularized version (3.1) of the dissipation function, so that
 323 the microstresses are given by (3.2). Equation (6.2) thus incorporates both microforce balance
 324 and the flow relation, and after discretization is equivalent to the smooth minimization problem
 325 (5.12). The magnitude of the perturbation η is chosen to be small enough for trends in the
 326 elastic-plastic behaviour to be captured with sufficient accuracy.

327 It has been shown in [20] that a sufficient condition for the existence of a unique solution to the
 328 purely dissipative problem is that there be some hardening present. Accordingly, we introduce
 329 a small amount of hardening to avoid pathologies in the numerical solutions; the hardening may
 330 be viewed as a small perturbation, which does not affect the overall features of the solutions.

Plastic incompressibility is enforced via the inclusion of the energy term

$$\psi_{\text{inc}} = \frac{\beta}{2}(\text{tr } \boldsymbol{\varepsilon}^p)^2,$$

331 whose derivative with respect to the vector \mathbf{p} is added to $R_{\mathbf{p}}$, where $\beta > 0$ is a penalty.

332 The problem is then one of solving equations (6.1) and (6.2) for the displacement \mathbf{u}_{n+1} and
 333 plastic strain increment $d\boldsymbol{\varepsilon}^p$.

The displacement and plastic strain fields (and their associated test functions) are approximated on a mesh of quadrilateral elements with conforming bilinear interpolations. The vector of global unknowns is denoted by $\mathbf{x} := (\mathbf{d}, \mathbf{p})$. A global Newton–Raphson procedure is used to linearize and iteratively solve the system of residual equations. An arbitrary variable evaluated at the current iteration (i) in time step $n + 1$ is denoted by $(\bullet)_{n+1}^{(i)} \equiv (\bullet)^{(i)}$. The linearized problem and the iterative update of the solution vector are then given by

$$\begin{aligned} \left. \frac{\partial \mathbf{R}}{\partial \mathbf{x}} \right|_{(i)} \cdot \Delta \mathbf{x} &= -\mathbf{R}^{(i)}, \\ \mathbf{x}^{(i+1)} &= \mathbf{x}^{(i)} + \Delta \mathbf{x}, \end{aligned}$$

334 where $\mathbf{R} := (R_{\mathbf{d}}, R_{\mathbf{p}})$.

335 The finite element library AceGen [18] is used to implement the finite element interpolation,
 336 and to compute the residual and tangent contributions using automatic differentiation. This ap-
 337 proach greatly simplifies the implementation. In addition, an adaptive time-stepping algorithm
 338 is employed.

339 We consider two examples, viz. a thin plate is subjected to biaxial deformation, and uniaxial
 340 extension of a rod.

341 **6.1 Biaxial deformation of a thin micro-plate**

342 In this example the role of the microscopic boundary conditions on the evolution of the problem
 343 are of particular interest. The material properties used in this example and the next are listed
 344 in Table 1, unless stated otherwise.

Table 1: Constitutive parameters used for the numerical examples unless stated otherwise

First Lamé parameter	λ	1.05×10^{-1}	$\text{N}/\mu\text{m}^2$
Poisson's ratio	ν	0.3	
Yield stress	Y	1×10^{-3}	$\text{N}/\mu\text{m}^2$
Regularization parameter	η	5×10^{-4}	
Incompressibility penalty	β	1×10^6	$\text{N}/\mu\text{m}^2$

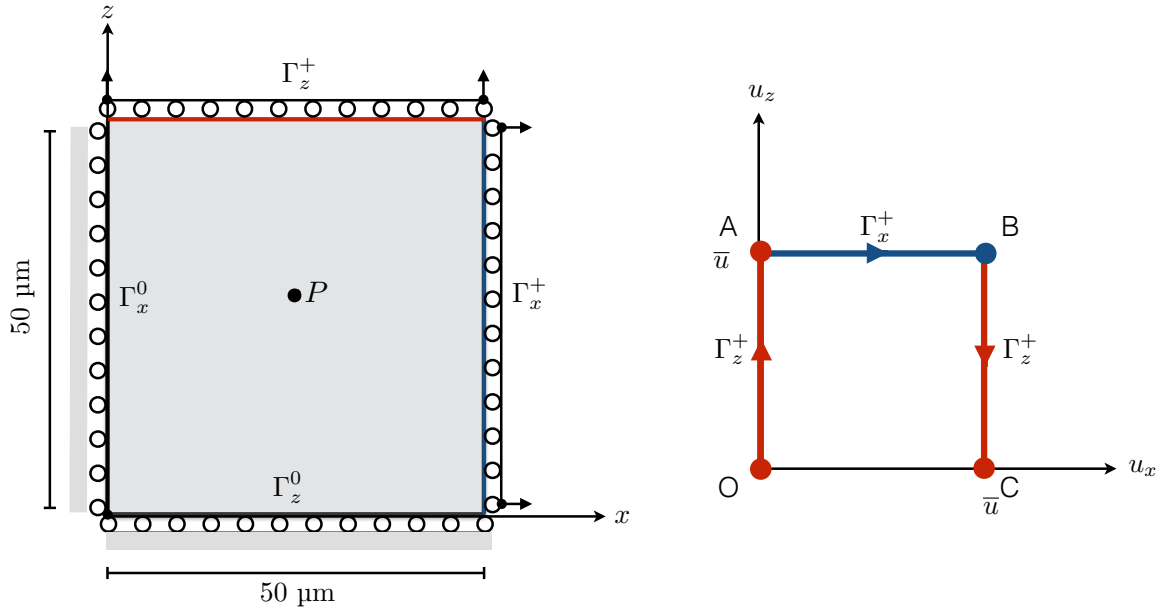


Figure 7: The problem of biaxial deformation in a plate and a schematic of the various stages of loading.

345 Consider the $50 \times 2 \times 50 \mu\text{m}$ plate shown in Fig. 7. The motion of the upper boundary Γ_z^+
 346 and the right boundary Γ_x^+ is defined in three stages. During load stage O–A (see Fig. 7), Γ_z^+
 347 is displaced in the positive z -direction by a distance $\bar{u} = 2/3 \mu\text{m}$ while Γ_x^+ is prevented from
 348 displacing in the x -direction. During load stage A–B, Γ_x^+ is displaced in the positive x -direction
 349 by a distance \bar{u} while Γ_z^+ is prevented from displacing in the z -direction. Finally, during stage
 350 B–C, the loading imposed during stage O–A is reversed by displacing Γ_z^+ a distance $-\bar{u}$ in the
 351 z -direction, while Γ_x^+ is prevented from displacing in the x -direction. Each loading stage (that

352 is, OA, AB, BC and CA) corresponds to a time of 0.5 s. The plate is free to displace in the
353 y -direction.

354 The domain is discretized using 2500 elements with one element through the thickness. The
355 maximum permissible time-step size is 1×10^{-3} s. The length scale is $l = 0.1L$, where $L = 50$
356 μm .

357 The influence of a microscopic boundary condition for the plastic strain evolution on the global
358 response is investigated by prescribing Γ_z^+ to be either (a) micro-free, (b) micro-hard from the
359 onset, or (c) micro-free for $0 \leq t < t_{\text{pass}}$, and thereafter preventing the evolution of plastic strain.
360 The boundary condition (c) is termed passivation. When imposed, passivation will occur at time
361 $t_{\text{pass}} = 0.4$ s. The response of a material point in the centre of the specimen $P = [25, 1, 25]$ μm
362 is monitored.

363 The evolution of the magnitude of the Cauchy stress and stress deviator at the point P for the
364 various microscopic boundary conditions is shown in Fig. 8. The macroscopic constraints give
365 rise to the volumetric contributions to the stress tensor. For the micro-free (a) and passivated
366 (c) boundary conditions, yield occurs at $t := t_Y \approx 0.38$ s when $|\text{dev } \boldsymbol{\sigma}| = Y$ (the initial yield
367 stress). The onset of yielding at point P is delayed when micro-hard boundary conditions are
368 imposed on the upper surface. Unsurprisingly, the presence of Dirichlet boundary conditions
369 on the plastic strain changes the global response as the residual expression (6.2) now contains
370 additional constraints. As seen in Fig. 8(b) the amount of hardening is minimal. The hardening
371 is linear for the micro-free problem but is more complex for the micro-hard and passivated
372 boundary conditions due to the contribution of the higher-order terms. Elastic unloading occurs
373 at the onset of load stage B–C.

374 The yield stress, that is, the stress at which global behaviour undergoes the transition from
375 elastic to elastic-plastic, is determined automatically as a result of the perturbed dissipation
376 function used in the computations. Thus this approach allows the yield stress to be obtained
377 despite a closed-form expression not being available, as discussed in Section 5.2.

378 For the passivation problem (c), the microscopic Dirichlet constraints on the plastic strain
379 evolution are imposed at $t_{\text{pass}} > t_Y$. The evolution of the stress state at point P shows that the
380 pre-passivation micro-free response transforms to a micro-hard response post-passivation. *This*
381 *transition occurs elastically.* This is the phenomenon is referred to in [8] as an elastic gap.
382 The system evolves to the curve corresponding to the micro-free boundary condition elastically,
383 which may be interpreted as the most efficient route.

384 The analysis in earlier sections does not allow for a closed-form expression for the yield sur-
385 face. This can however be explored numerically with reference to the stress response in σ_{11} – σ_{33}
386 space, as shown in Fig. 9. Although not indicated, the micro-free response is identical to that
387 obtained using a classical return mapping algorithm (closest-point projection) at the level of the

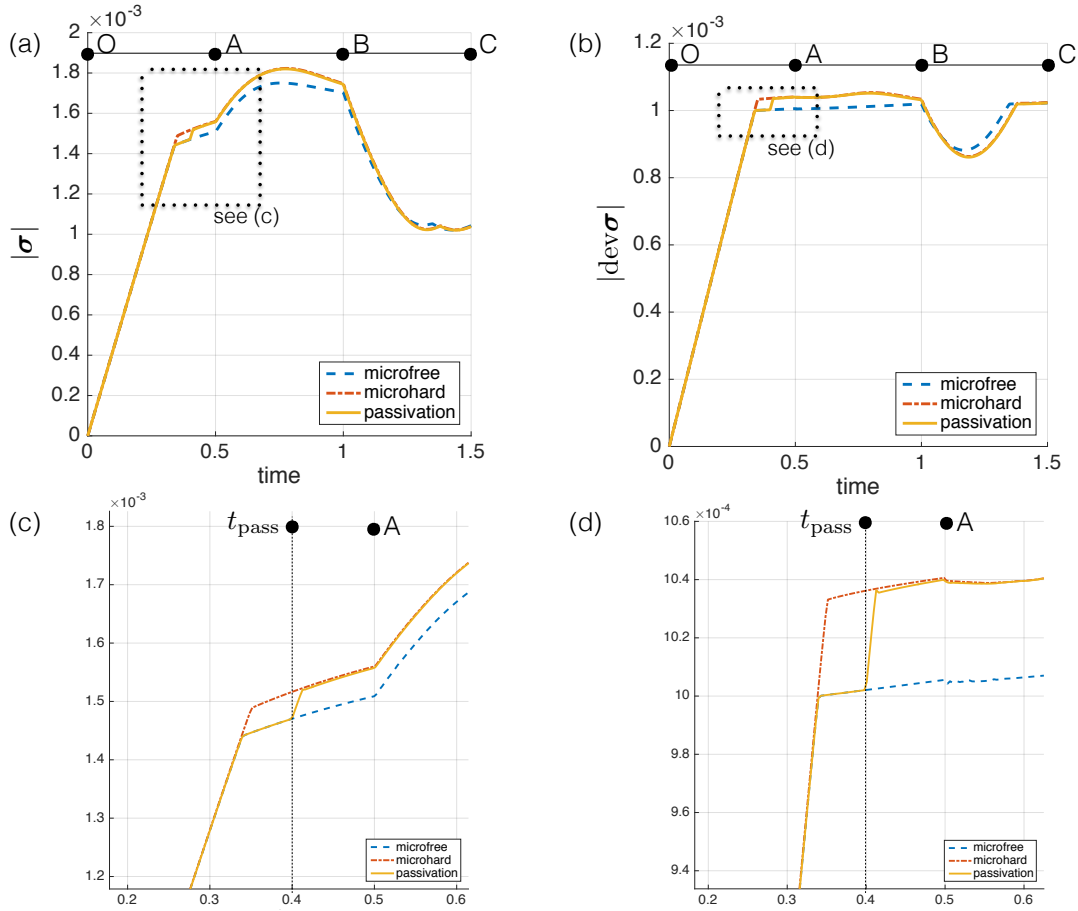


Figure 8: The evolution of (a) the Cauchy stress and (b) the Cauchy stress deviator at point P for the various boundary conditions. The various load stages are also indicated.

388 integration point for the non-gradient, rate-independent J_2 plasticity problem. This confirms
 389 that the global formulation based on a micro-force balance is essentially equivalent to the local
 390 formulation in the absence of gradients. It also confirms that for the micro-free condition the
 391 choice of a primal formulation with the dissipation function D_2 is equivalent to the dual problem
 392 with a von Mises yield surface. The yield surface for the micro-free problem in Fig. 9 can thus
 393 be seen as the von Mises yield surface corresponding to the classical problem.

394 The yield surface for the micro-hard problem is expanded relative to the micro-free one, con-
 395 sistent with the elastic gap transition reported earlier. Furthermore, due to the gradient
 396 contributions to the hardening, the expansion is not uniform. The yield surface for the passi-
 397 vated problem is on the micro-free surface until $t = t_{\text{pass}}$, after which it moves elastically to the
 398 micro-hard one.

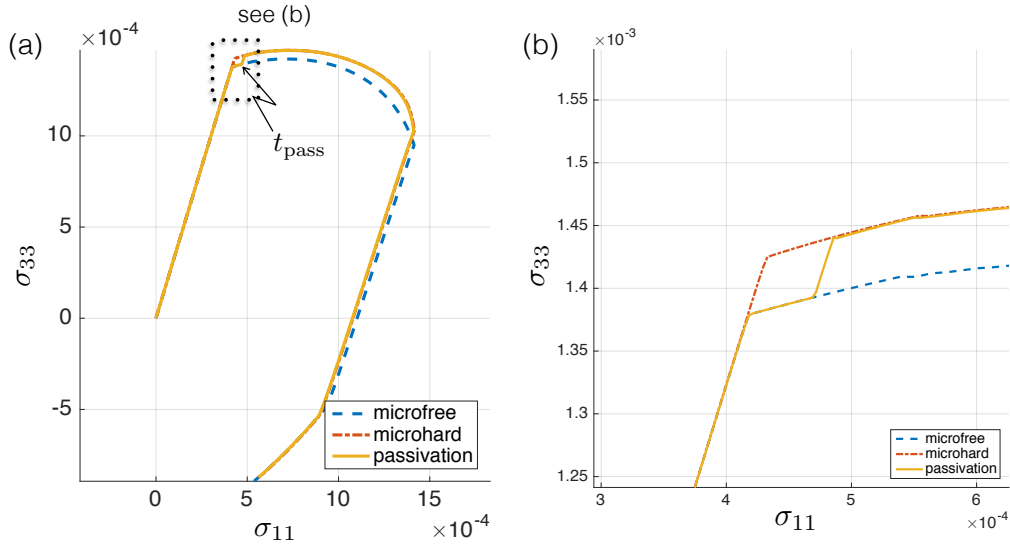


Figure 9: The evolution of the Cauchy stress components σ_{11} and σ_{33} for the various boundary conditions.

6.2 Extension of a micro-rod

Consider a rod having radius $25 \mu\text{m}$ and length $L = 50 \mu\text{m}$, and subjected to a prescribed displacement in the axial direction of $0.5 \mu\text{m}$ applied to the upper and lower faces with normals \mathbf{e}_3 and $-\mathbf{e}_3$, respectively. The curved side is traction-free. Due to symmetry, only the upper quarter of the rod is modelled as shown in Fig. 10. The prescribed displacement is imposed incrementally over 0.5 s . The response of the system at a material point labelled A and located at $[0, 0, 12.5] \mu\text{m}$ is recorded. The length scale, unless otherwise stated, is $l = 0.2L$. The domain is discretized using 6527 elements.

As in the previous example, the consequences of choosing different microscopic boundary conditions on the upper boundary of the domain, denoted Γ_u , are investigated. The curved side is at all times micro-traction free. Passivation occurs at $t_{\text{pass}} = 0.25 \text{ s}$ which is well into the plastic range.

The response at point A for the various choices of the microscopic boundary conditions on Γ_u is shown in Fig. 11. The relation between the magnitudes of the Cauchy stress and the strain, shown in Fig. 11(a), clearly contain the same features discussed in the previous example: an increase in the perceived yield strength for the micro-hard condition and an elastic gap for the passivation problem. Furthermore, the size of the elastic gap increases with increasing length scale. This relation between the size of the elastic gap and the length scale was also observed in [8].

The evolution of the quantity $\phi := |\mathbf{S}|/Y$, which corresponds to the classical yield function, is

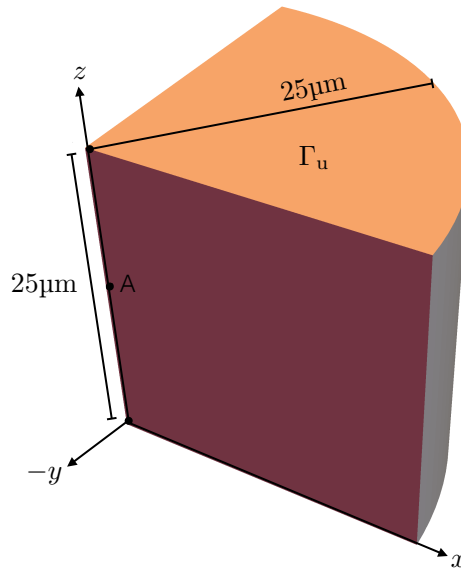


Figure 10: Computational domain for the problem of the extension of a rod.

419 shown in Fig. 11(b). As expected, ϕ is in the range $0 < \phi < 1$ in the elastic region, and $\phi = 1$
 420 during plastic flow for all microscopic boundary conditions. The elastic gap at $t = t_{\text{pass}} = 0.25$
 421 s is also clearly indicated for the passivation problems as ϕ drops below unity.

422 The evolution of the stress in σ_{11} - σ_{22} - σ_{33} space is shown in Fig. 11(c). For the micro-free
 423 condition, the stress state is uniaxial with σ_{33} the only non-zero stress component. The stress
 424 state remains at the point on the yield surface where initial yield occurred. The stress state is
 425 spatially uniform throughout the specimen and there are no plastic strain gradients present.

426 For the micro-hard boundary condition the stress evolves symmetrically in the σ_{11} and σ_{22}
 427 directions post yield. The micro-hard boundary condition constrains all components of the
 428 plastic strain, thereby inducing a stress response in directions other than that of the loading.
 429 Microscopic Dirichlet conditions on the plastic strain result in plastic strain gradients and Cauchy
 430 stresses in directions other than the loading direction. The yield surface for the passivation
 431 problem is identical to that of the micro-free problem prior to passivation. The stress state
 432 at the point of passivation will have only a σ_{33} component. Upon passivation, the elastic gap
 433 occurs. For the larger length scale of $l = 0.4L$ the yield surface appears to increase above the
 434 micro-hard one. It should be noted that post-passivation, the stress components plotted are no
 435 longer the principal stress components.

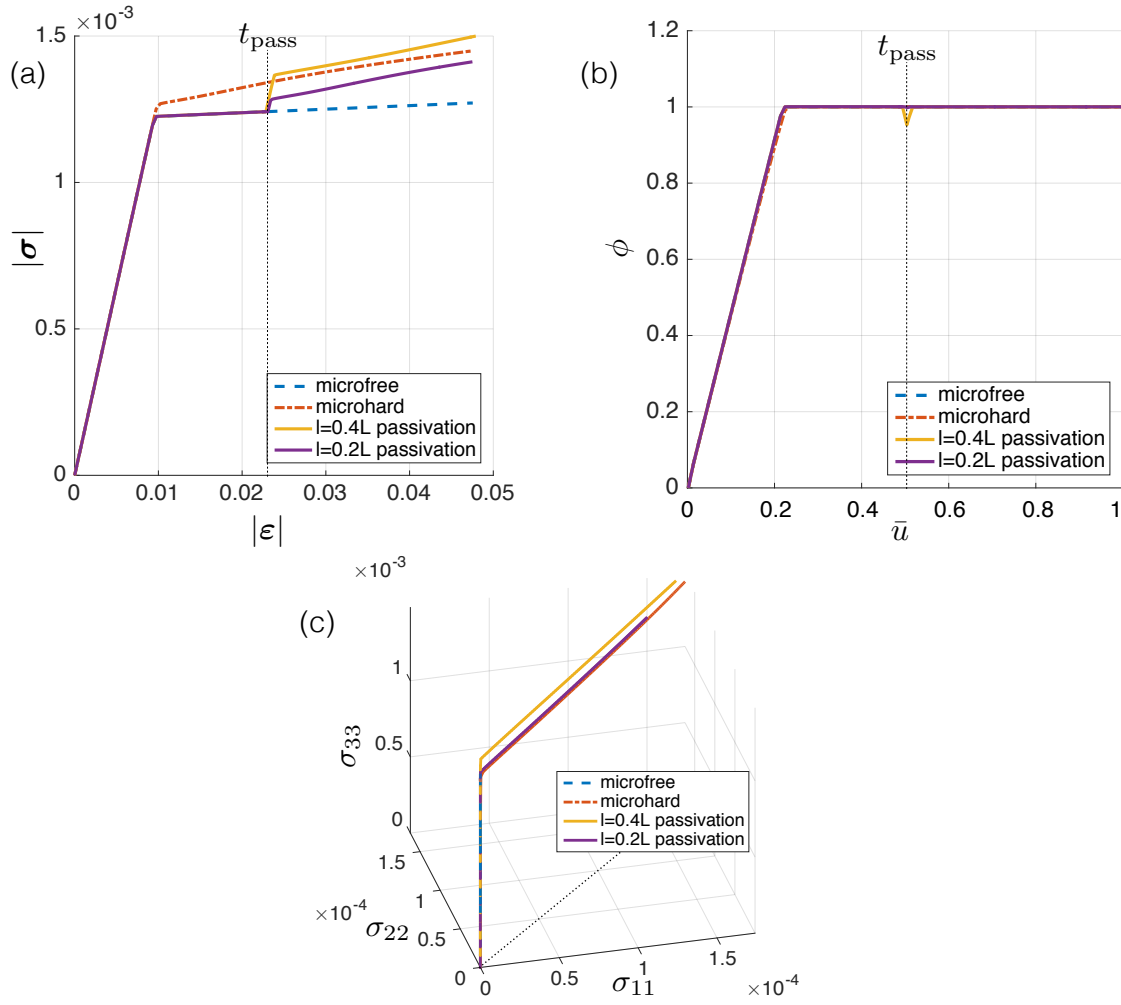


Figure 11: The evolution of the state at material point A for the rod extension problem for various microscopic boundary conditions and different length scales. The relation between $|\boldsymbol{\sigma}|$ and $|\boldsymbol{\varepsilon}|$ is shown in (a). The evolution of the canonical yield function ϕ and the Cauchy stress are shown in (b) and (c), respectively. The curve corresponding to the micro-hard boundary condition is for the length scale $\ell = 0.2L$.

7 Concluding remarks

436

437 A theoretical and computational investigation has been carried out of a dissipative model of
 438 rate-independent strain-gradient plasticity, that is, one in which gradient terms are accounted
 439 for only in the flow relation. The global nature of the flow relation, previously reported in [20],
 440 is reiterated. The most appropriate and effective approach to formulating the flow relation is
 441 through the use of a dissipation function; this form of the relation is especially useful in the
 442 context of numerical investigations. Dual formulations in terms of the yield function and a

443 normality relation have been approached using the tools of convex analysis. It is not possible,
444 using conventional tools, to invert the flow relation to obtain the yield surface corresponding to
445 the global dissipation function. This objective has been investigated further in the context of
446 the fully discrete problem, for which an upper bound to the elastic region is found.

447 The numerical investigation casts further light on the response using the dissipative theory
448 in situations of non-proportional loading due to passivation. Post-yield behaviour has been
449 investigated. The elastic gap reported in [8] and analyzed in [12] has been observed in situations
450 in which passivation has been imposed. It has been possible to interpret the gap mathematically,
451 using the expression for the yield function as a maximum, taken over all admissible plastic
452 strain increments, of a function involving the dissipation: the vector of admissible increments
453 is necessarily smaller in dimension following passivation, and the corresponding maximum may
454 therefore be smaller than that in the step preceding passivation. The elastic gap has also
455 been observed to constitute an “efficient” transition from a stress-strain curve corresponding
456 to a micro-free boundary condition, to that which is obtained assuming micro-hard boundary
457 conditions.

458 The dissipative model of strain-gradient plasticity is mathematically well posed [20]. The pres-
459 ence of the elastic gap, at least in the case of a change of boundary conditions achieved through
460 passivation, is indisputable from theoretical and numerical perspectives, as has been demon-
461 strated here and in the works by Fleck, Hutchinson and Willis. The actual physical response
462 is one that awaits experimental investigation, as has been emphasized by these authors. Such
463 testing would clarify the predictive nature of the theory: for example, as suggested in [12], as a
464 theory in which small plastic deformation is approximated as zero plastic deformation.

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470 **References**

- 471 [1] Aifantis, E.C. (1984). On the microstructural origin of certain inelastic models. *J. Engng*
472 *Mat.Tech.* **106** 326–330.
- 473 [2] Anand, L., Gurtin, M.E., Lele, S.P. and Gething, C. (2005). A one-dimensional theory of
474 strain-gradient plasticity: Formulation, analysis, numerical results. *J. Mech. Phys. Solids*

- 476 [3] Bardella, L. and Panteghini, A. (2015). Modelling the torsion of thin metal wires by distortion
477 gradient plasticity. *J. Mech. Phys. Solids* **78** 467–492.
- 478 [4] Chiricotto, M. and Giacomelli, L. and Tomassetti, G. (2016). Dissipative scale effects in
479 strain-gradient plasticity: the case of simple shear. *SIAM J. Appl. Math.* **76**(2) 688–704.
- 480 [5] Ekeland, I. and Temam, R. (1976). *Convex Analysis and Variational Problems*. North-
481 Holland, Amsterdam.
- 482 [6] Evans, A.G. and Hutchinson, J.W. (2009). A critical assessment of theories of strain gradient
483 plasticity. *Acta Materialia* **57** 1675–1688.
- 484 [7] Fleck, N.A. and Hutchinson, J.W. (2001). A reformulation of strain gradient plasticity. *J.*
485 *Mech. Phys. Solids* **49** 2245–2271.
- 486 [8] Fleck, N.A., Hutchinson, J.W. and Willis, J.R. (2015). Strain-gradient plasticity under non-
487 proportional loading. *Proc. R. Soc A* **470** 20140267.
- 488 [9] Fleck, N.A., Hutchinson, J.W. and Willis, J.R. (2015). Guidelines for constructing strain
489 gradient plasticity theories. *J. Appl. Mech.* **82** 071002-1-10.
- 490 [10] Fleck, N.A. and Willis, J.R. (2009). A mathematical basis for strain-gradient plasticity -
491 Part I: Scalar plastic multiplier. *J. Mech. Phys. Solids* **57** 151–177.
- 492 [11] Fleck, N.A. and Willis, J.R. (2009). A mathematical basis for strain-gradient plasticity -
493 Part II: Tensorial plastic multiplier. *J. Mech. Phys. Solids* **57** 1045–1057.
- 494 [12] Fleck, N.A. and Willis, J.R. (2015). Strain gradient plasticity: energetic or dissipative?
495 *Acta Mech. Sinica* **31** 465–472.
- 496 [13] Gao H., Huang Y. and Nix W.D. 1999. Mechanism-based strain gradient plasticity ? I.
497 Theory. *J. Mech. Phys. Solids* **47** 1239–1263.
- 498 [14] Gao H., Huang Y., Nix W.D. and Hutchinson J.W. (1999). Modeling plasticity at the
499 micrometer scale, *Naturwissenschaften* **86** 507–515.
- 500 [15] Gudmundson, P. (2004). A unified treatment of strain gradient plasticity. *J. Mech. Phys.*
501 *Solids* **52** 1379–1406.
- 502 [16] Gurtin, M.E., Anand, L. (2005). A theory of strain-gradient plasticity for isotropic, plasti-
503 cally irrotational materials. Part I: small deformations. *J. Mech. Phys. Solids* **53** 1624–1649.
- 504 [17] Han, W. and Reddy, B.D. (2013). *Plasticity: Mathematical Theory and Numerical Analysis*.
505 Springer, New York and Berlin.

- 506 [18] Korelc, J. (2002). Multi-language and multi-environment generation of nonlinear finite ele-
507 ment codes. *Engng Comp.* **18**(4) 312–327.
- 508 [19] Panteghini, A. and Bardella, L. (2016). On the Finite Element implementation of higher-
509 order gradient plasticity, with focus on theories based on plastic distortion incompatibility.
510 *Comput. Methods Appl. Mech. Engrg.* **310** 840–865.
- 511 [20] Reddy, B.D. (2011). The role of dissipation and defect energy in variational formulations
512 of problems in strain-gradient plasticity. Part 2: single-crystal plasticity. *Continuum Mech.*
513 *Thermodyn.* **23** 547–549.
- 514 [21] Reddy, B.D., Ebobisse, F. and McBride, A.T (2008). Well-posedness of a model of strain
515 gradient plasticity for plastically irrotational materials. *Int. J. Plast.* **24** 55–73.
- 516 [22] Rockafellar, R.T. (1970). *Convex Analysis*. Princeton University Press, Princeton, N.J.